

# Investigation of the relations between structure and microwave dielectric properties of divalent metal tungstate compounds

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## Abstract

The microwave dielectric characteristics of  $AWO_4$  ( $A = \text{Mg, Zn, Mn, Ca, Sr, Ba}$ ) compounds and their relations with structure have been investigated as a function of A cations. Crystal structure was investigated by XRD; the structure changed from wolframite to scheelite with increasing size of A cation. Dielectric properties such as permittivity, temperature coefficient of resonant frequency ( $\tau_f$ ), and quality factors ( $Q \times f$ ) were found to be correlated with the size of A cations and the structure of compounds. Permittivity decreased with the size of A cations, and temperature coefficient of resonant frequency was correlated with unit cell volume.  $AWO_4$  compounds are suitable for the applications of low- $k$  dielectric materials; permittivity = 8–15,  $\tau_f = -50$  to  $-70$ , and  $Q \times f = 60,000$ –70,000.

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## 1. Introduction

According to the rapid progress in mobile telecommunication technologies, such as mobile phones, high quality microwave dielectric ceramics are strongly required. Ceramics using as a microwave substrate application, should be have low permittivity, high quality factor, and low temperature coefficient of resonant frequency.  $Al_2O_3$ , rare-earth aluminates and phosphates have been investigated for using this purpose. Although these compounds have good dielectric properties, they have some demerits such as high sintering temperature. Therefore there are attempts to search for new materials with better microwave dielectric properties.<sup>1–3</sup>

The major natural tungsten ores are formed by the minerals scheelite,  $(\text{Ca, Sr, Ba})WO_4$  and wolframite  $(\text{Mg, Zn, Mn, Fe})WO_4$ . Depending on the size of the cation, tungstates can crystallize in either the scheelite or the wolframite structure. Scheelite structures with the tungsten in tetrahedral coordination are formed with large bivalent cations like Ca, Ba, Pb and Sr. Alternatively, with smaller bivalent cations (Fe, Mn, Co, Ni, Mg, Zn) the wolframite structure will be formed with the

tungsten in octahedral coordination.<sup>4</sup>  $AWO_4$  compounds have attracted much interest in just their use as scintillating medium and in electro-optic application. For application of microwave dielectric materials, little research has been done.<sup>5,6</sup> So, for investigation of possibility of microwave application, and relations between  $AWO_4$  structure and dielectric properties, we researched the divalent metal tungstate compounds.

## 2. Experimental procedure

The starting materials used were  $WO_3$  (high purity),  $ACO_3$  ( $A = \text{Ca, Sr, Ba}$ ),  $Mn_3O_4$  and  $AO$  ( $A = \text{Mg, Zn}$ ). The powders were mixed for 24 h in a ball mill with  $ZrO_2$  media using ethanol. The mixed slurry was dried and then calcined at  $800^\circ\text{C}$  for 2 h in air. The calcined powders were ground for 24 h in a ball mill. After drying milled powders, the dried powders were pressed into a pellet and pellets were sintered in the temperature range of  $1000$ – $1150^\circ\text{C}$  for 2 h at heating rates of  $5^\circ\text{C}/\text{min}$ . The bulk density of the sintered samples was measured by the Archimedes' method. Structure of calcined powders and sintered samples was examined by X-ray diffraction (M18XHF, Macscience Instruments, Japan). Polished surfaces of sintered samples were thermally etched in order to examine their microstructures using a scanning electron microscopy (SEM, JEOL-6330F, JEOL, Japan). The quality factor, the dielectric constant, and the temperature coefficient of resonant frequency

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of the sintered specimens were measured with a HP 8720 network analyzer (Model HP8720C, Hewlett Packard, USA).

### 3. Results

Figs. 1 and 2 shows the X-ray diffraction of calcined AWO<sub>4</sub> compounds. The single-phase structure of all compounds was confirmed by X-ray powder diffraction. Diffraction peaks shifted to lower angle in the same structure as the radius of cation increased. Structure of AWO<sub>4</sub> changed from wolframite structure (A = Mg, Mn, Zn) to scheelite structure (A = Ca, Sr, Ba). Wolframite structure is monoclinic and scheelite is tetragonal.

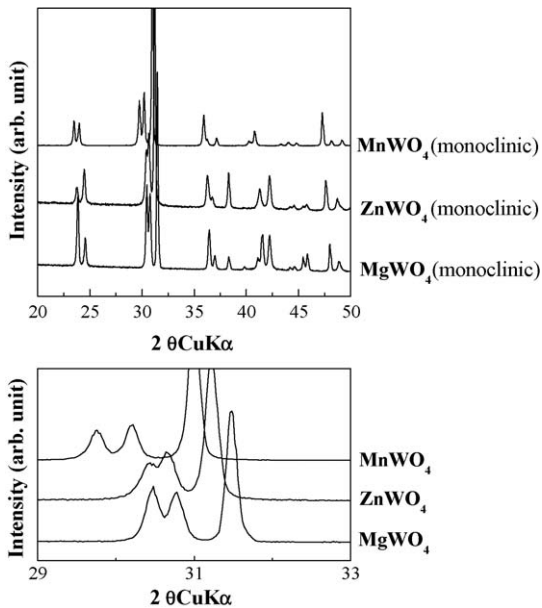


Fig. 1. X-ray diffraction patterns of calcined AWO<sub>4</sub> powders (A = Mg, Zn, Mn).

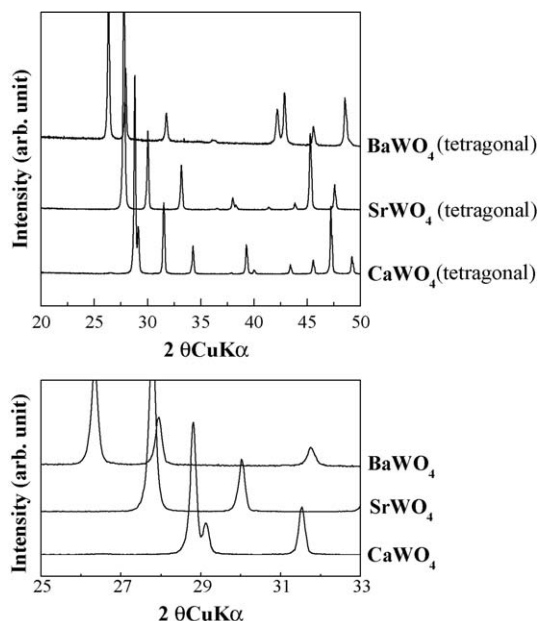


Fig. 2. X-ray diffraction patterns of calcined AWO<sub>4</sub> powders (A = Ca, Sr, Ba).

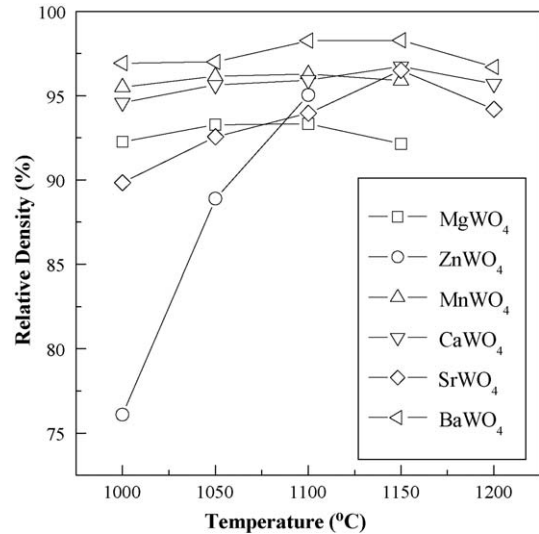


Fig. 3. Bulk density of AWO<sub>4</sub> as a function of sintering temperature.

Table 1  
Dielectric properties of AWO<sub>4</sub> compounds

Material	Sintering temperature (°C)	$\epsilon_r$	$Q \times f$ (GHz)	$\tau_f$ (ppm/°C)	Porosity
MgWO <sub>4</sub>	1050	13.5	69,000	-58	0.067
ZnWO <sub>4</sub>	1100	17.6	65,000	-60	0.050
MnWO <sub>4</sub>	1100	14.8	32,000	-64	0.037
CaWO <sub>4</sub>	1150	10.4	63,000	-53	0.033
SrWO <sub>4</sub>	1150	8.1	56,000	-55	0.035
BaWO <sub>4</sub>	1150	8.1	57,500	-78	0.017

Fig. 3 shows the bulk relative density of AWO<sub>4</sub> samples. The compounds of AWO<sub>4</sub> could be sintered into dense bodies at the temperature about 1150 °C. Most of the compounds have sintered densities more than 95% of their theoretical densities. The samples of MgWO<sub>4</sub>, ZnWO<sub>4</sub>, MnWO<sub>4</sub> were melted at 1200 °C, 1150 °C, 1200 °C, respectively.

The SEM micrographs of the surfaces of AWO<sub>4</sub> samples are shown in Fig. 4. Grain size was 3–5 μm and the dense microstructure was confirmed by a SEM.

The dielectric properties of AWO<sub>4</sub> compounds were summarized in Table 1. The samples of scheelite structure had good microwave dielectric properties. In the case of CaWO<sub>4</sub>,  $Q \times f$  was about 60,000 and  $\epsilon_r$  about 10, and  $\tau_f$  about -55 ppm/°C. The compounds of wolframite had higher permittivity than that of scheelite structure.

### 4. Discussion

Shannon<sup>7</sup> suggested that molecular polarizabilities of complex substances can be broken up into the polarizabilities of constituent ions. In the case of AWO<sub>4</sub>, molecular polarizability can estimate from following equation.

$$\alpha(\text{AWO}_4) = \alpha(\text{A}^{2+}) + \alpha(\text{W}^{6+}) + 4\alpha(\text{O}^{2-}) \quad (1)$$

where  $\alpha$  is polarizability. The dielectric constants of AWO<sub>4</sub> compounds could be calculated with the polarizability from

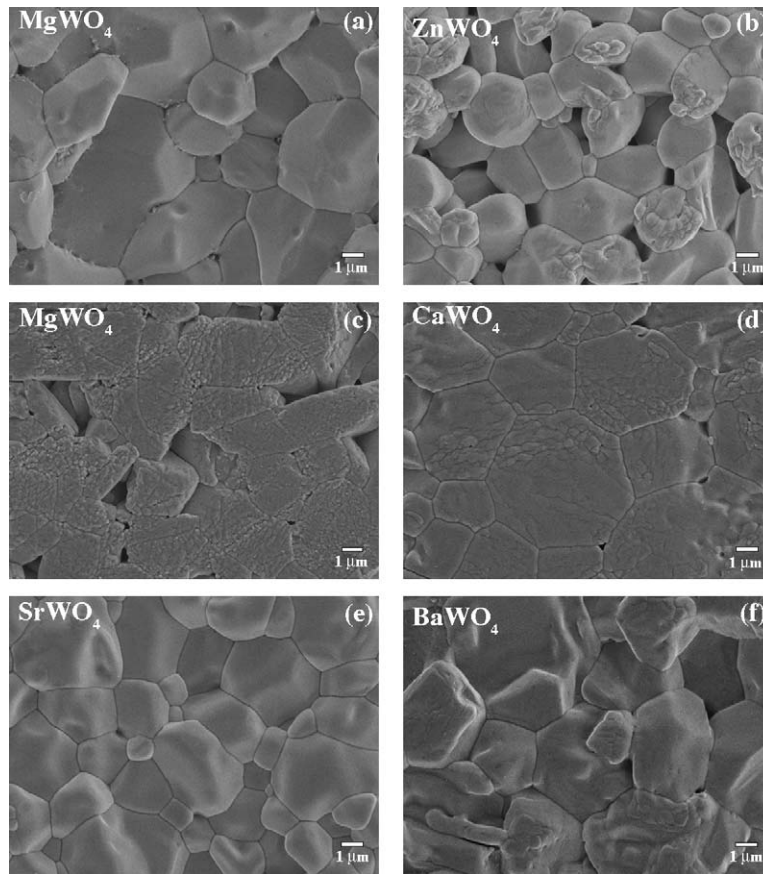


Fig. 4. SEM micrographs of sintered sample of AWO<sub>4</sub>: (a) MgWO<sub>4</sub> sintered at 1050 °C for 2 h, (b) ZnWO<sub>4</sub> sintered at 1100 °C for 2 h, (c) MnWO<sub>4</sub> sintered at 1100 °C for 2 h, (d) CaWO<sub>4</sub> sintered at 1150 °C for 2 h, (e) SrWO<sub>4</sub> sintered at 1150 °C and (f) BaWO<sub>4</sub> sintered at 1150 °C for 2 h.

Shannon's suggestion and Clausius–Mosotti relation,<sup>7</sup>

$$\varepsilon_s = \frac{3V_m + 8\pi\alpha_D}{3V_m - 4\pi\alpha_D} \quad (2)$$

where  $V_m$  is molar volume. Molar volume of AWO<sub>4</sub> can be calculated with the unit cell parameters determined from the diffraction pattern, and the values of molar volume are shown in Table 2. And we calculated the dielectric constants of AWO<sub>4</sub> with 3.2 for the mean value of polarizability of W<sup>6+</sup> on Shannon's suggestion.<sup>7</sup> The permittivities of AWO<sub>4</sub> compounds are shown in Table 2. The permittivities in the Table 2 were corrected by following equation.

$$\varepsilon_r = \varepsilon^{\text{obs}}(1 + 1.5P), \quad P = 1 - \frac{\rho}{\rho_{\text{th}}} \quad (3)$$

Table 2  
Polarizability, molar volume, calculated and observed permittivity of AWO<sub>4</sub> compounds

Material	$\alpha_D$	$V_m$	$\varepsilon_s^{\text{cal}}$	$\varepsilon_r$
MgWO <sub>4</sub>	12.56	65.55	13.2	13.5
ZnWO <sub>4</sub>	13.28	65.50	16.3	17.6
MnWO <sub>4</sub>	13.88	69.50	16.3	14.8
CaWO <sub>4</sub>	14.4	78.13	11.1	10.4
SrWO <sub>4</sub>	15.48	87.83	9.5	8.1
BaWO <sub>4</sub>	17.64	100.25	9.4	8.1

where  $P$  is porosity.<sup>9</sup> The calculated and observed permittivities showed good agreement. The wolframite structure has hexagonally close-packed oxygens with certain octahedral sites filled by A and W cations in an ordered way, on the other hands, the scheelite structure has cubic close-packed array of A<sup>2+</sup> and [WO<sub>4</sub>]<sup>2-</sup> units with the coordination numbers of 8 and 4 oxygen atoms for the A and W cations, respectively. The atmos Ba, Ca and Sr had larger polarizabilities than the atoms of Mg, Zn, and Mn. However, the dielectric constants of wolframite structure were larger than those of scheelite structure. This can be explained by Clausius–Mosotti equation. As Clausius–Mosotti equation, polarizability and molar volume determine the dielectric constant. And as molar volume increase, dielectric constant decrease rapidly.<sup>7</sup> Therefore, compounds of scheelite structure have smaller dielectric constants in spite of larger polarizabilities by their larger molar volumes.

Fig. 5 shows relation between the temperature coefficient of permittivity,  $\tau_\varepsilon$  and unit-cell volume of AWO<sub>4</sub>. Generally temperature coefficient of resonant frequency,  $\tau_f$  is represented by the thermal expansion coefficient,  $\alpha_1$  and the temperature coefficient of permittivity,  $\tau_\varepsilon$  as follows:

$$\tau_f = -\alpha_1 - \frac{1}{2}\tau_\varepsilon \quad (4)$$

The thermal expansion coefficient of ceramic is generally in the range of 10 ppm/°C, therefore the sign and magnitude of

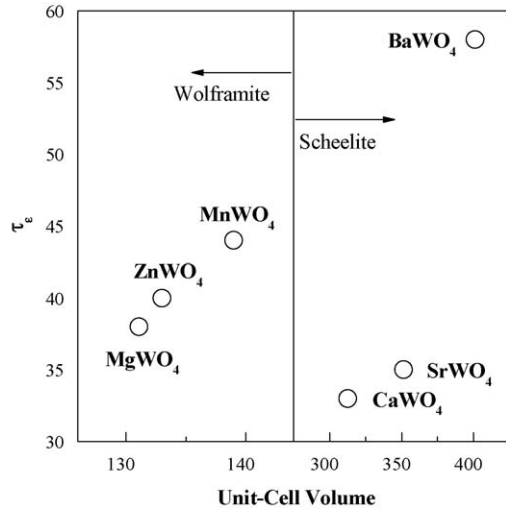


Fig. 5. Temperature coefficient of permittivity of AWO<sub>4</sub> compounds as a function of unit-cell volume.

$\tau_f$  depends on  $\tau_\varepsilon$ . From Clausius–Mosotti equation,  $\tau_\varepsilon$  can be derived a following expression.

$$\tau_\varepsilon = \frac{1}{\varepsilon} \left( \frac{\partial \varepsilon}{\partial T} \right) = \frac{(\varepsilon - 1)(\varepsilon + 2)}{\varepsilon} (A + B + C),$$

$$\left( A = -\frac{1}{3V} \left( \frac{\partial V}{\partial T} \right)_P, \quad B = \frac{1}{3\alpha_m} \left( \frac{\partial \alpha_m}{\partial V} \right)_T \left( \frac{\partial V}{\partial T} \right)_P, \right.$$

$$\left. C = \frac{1}{3\alpha_m} \left( \frac{\partial \alpha_m}{\partial T} \right)_V \right) \quad (5)$$

Bosman and Havinga suggested that the terms  $A$  and  $B$ , which correlate to volume expansion, have very similar in magnitude but opposite in sign, therefore the resulting effect at  $\tau_\varepsilon$  of terms  $A$  and  $B$  is generally ignored.<sup>8–10</sup> Lee<sup>9</sup> suggested that the term  $C$  represents the restoring force acting on the ions and restoring force depends on the shape of potential well, therefore this term correlated with structure and lattice parameter. Since the unit-cell volume has strong correlation with lattice energy,  $\tau_\varepsilon$  is proportional to the relative magnitude of unit-cell volume. As can be seen in Fig. 5 the magnitude of  $\tau_\varepsilon$  increased as the unit-cell volume increased in the same structure of AWO<sub>4</sub>.

## 5. Conclusions

The dielectric properties of AWO<sub>4</sub> have been investigated. AWO<sub>4</sub> are feasible as microwave dielectric materials due to their good dielectric properties, permittivity is 8–17, quality factor is 32,000–69,000, and  $\tau_f$  is –53 to –78. The changes in permittivities of AWO<sub>4</sub> compounds were described in terms of the difference in ionic polarizability of A cations and molar volume. The temperature coefficients of resonant frequency were shown to be correlated to the unit-cell volume of AWO<sub>4</sub> for both wolframite and scheelite structures.

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